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1	Crystalline and porous CoSe dendrimeric architectures for efficient Oxygen
2	Evolution Reaction
3	Muhammad Bilal ^a , Rashid ^a , Amna Altaf ^a , Nadeem Baig ^b , Ghayoor Abbas Chotana ^c , Raja
4	Shahid Ashraf ^d , Shahid Rasool ^e , Ayman Nafadi ^f , Manzar Sohail ^{a*}
5	^a Department of Chemistry, School of Natural Sciences, National University of Sciences and
6	Technology, H-12, Islamabad 44000, Pakistan. Email: <u>manzar.sohail@sns.nust.edu.pk</u>
7	^b Interdisciplinary Research Center for Membranes and Water Security, King Fahd University
8	of Petroleum and Minerals, Dhahran, 31261 Saudi Arabia.
9	^c Department of Chemistry and Chemical Engineering, Syed Babar Ali School of Science and
10	Engineering, Lahore University of Management Sciences, Lahore 54792, Pakistan.
11	^d Department of Chemistry, Government College University Lahore 54000, Pakistan.
12	^e School of Engineering, Newcastle University, Newcastle Upon Tyne NE1 7RU, U.K and
13	Faculty of Engineering and Environment, Northumbria University, Newcastle Upon Tyne NE1
14	8ST, U.K.
15	^f Department of Chemistry, College of Science, King Saud University, Riyadh, 11451, Saudi
16	Arabia
17	
18	
19	* Corressponding Author. Department of Chemistry, School of Natural Sciences, National
20	University of Sciences and Technology, H-12, Islamabad 44000, Pakistan. Email:
21	manzar.sohail@sns.nust.edu.pk
22	

25 Abstract

Development of efficient and economical electrocatalyst for oxygen evolution reaction is the 26 key challenge to renewable energy technologies. Metal selenides are attractive candidates for 27 electrocatalytic water oxidation because of suitable surface-active sites they provide for the 28 29 reaction. Herein, we report the preparation of less explored hexagonal cobalt selenide (CoSe) 30 for oxygen evolution reaction through facile and environmentally benign one-step hydrothermal method. Reaction conditions were precisely tailored for the development of 31 32 highly crystalline and porous dendrimeric architectures of CoSe. Owing to its exclusive porous and the dendrimeric crystalline network and large electrochemical surface area, the superior 33 CoSe electrocatalyst (that is 16H) showed excellent electrochemical activity with remarkably 34 35 low overpotential (250 mV at 10 mA cm⁻²) and very high current density (570 mA cm⁻²) in a small potential window. The Tafel slope of 16H sample was 56 mV dec⁻¹ which indicates the 36 faster kinetics at the catalyst surfaces. Moreover, it also showed excellent stability under harsh 37 38 oxidative condition in a 24-hour long stability test experiment.

39 Keywords

40 CoSe; electrocatalyst; oxygen evolution reaction; water splitting; renewable energy

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50 **1. Introduction**

Electrochemical water splitting is one of the most promising approaches of converting 51 electrical energy into chemical energy in hydrogen fuel¹⁻³. Water splitting reaction includes 52 53 two half reactions: hydrogen evolution reaction (HER) and oxygen evolution reaction (OER). However, a major challenging reaction in water splitting is the oxygen evolution reaction 54 (OER). OER is kinetically sluggish due to multiple-electron transfer in complicated O-H bond 55 breakings and subsequent O=O bond formations, which require a high overpotential⁴⁻⁸. An 56 efficient electrocatalyst is required for reducing energy barrier and enhancing energy 57 conversion efficiency 9. Pioneering studies for the OER have revealed that RuO₂ and IrO₂ are 58 promising electrocatalysts. However, their high cost and scarcity severely impede their 59 widespread applications¹⁰⁻¹³. This limitation encouraged researchers to explore efficient and 60 economic electrocatalysts for the water oxidation reaction. The reported electrocatalysts for 61 OER include transition metal oxides, hydroxides, selenides, phosphides, nitrides, and 62 perovskites¹⁴⁻¹⁹. Among all these transition metal chalogenide is have gained attention owing to 63 their extraordinary chemical and physical properties, quantum size, luminescence and optical 64 properties ²⁰. From the various transition metal selenides and their composites for the OER, 65 cobalt selenides have received significant attention due to their low cost, earth abundance, 66 chemical stabilities, and comparable efficiencies to that of IrO₂ and RuO₂²¹⁻³⁷. Till now, cobalt-67 based catalyst, particularly cobalt selenides and oxides have gained attention owing to their 68 higher catalytic activity and corrosion stability. To enhance the efficiency of Cobalt-based 69

70 electrocatalyst, synthetic strategies has been varied, doping and composites have also been 71 synthesized. Although, pure cobalt selenide is found a benign catalyst for oxygen evolution reaction ³⁸. The efficient electrocatalytic activity of cobalt selenides is due to suitable surface-72 73 active sites for water splitting. Stoichiometric CoSe₂ is the most reported layered cobalt selenide for oxygen evolution reaction^{21, 39-42}. However, there only a few reports in literature 74 for OER with other pure phases of cobalt selenides. These include Co_{0.85}Se, Co₃Se₄, Co₇Se₈, 75 Co₉Se₈ and CoSe^{40, 43-49}. The OER electrocatalytic activities of all these pure cobalt selenide 76 phases including the current work are given in supplimentry table 1. Highly crystalline and 77 porous architectures of suitable materials are important to expose maximum surface active sites 78 79 for reactants and to enhance the electron transfer process during a reaction. Selenium is sensitive to the applied reaction conditions because it exists in multiple oxidation states. 80 Physical properties of transition metal dichalcogenides are strongly influenced by metal cations 81 d-electronic configuration. In CoSe exhibits +2 charge (d_7) and it is paramagnetic in nature and 82 this metallic nature enhance the performance of CoSe ⁵⁰. 83

Therefore, in the current study, less explored stochiometric CoSe architectures are prepared under varying reaction conditions through environmentally benign facile one-pot hydrothermal method. The reaction conditions are optimized to produce highly crystalline, and porous CoSe architectures. Also, their detailed electrochemistry for water oxidation is investigated.

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2. Experimental

Cobalt chloride hexa-hydrate, Selenium powder, hydrazine hydrate, were purchased from
Sigma Aldrich and were used without further purification. Deionized water obtainbed from
Milli-Q (EQ-7000) direct water purification system (Merck KGaA, Darmstadt 64297,
Germany), was used for making all the solutions.

93 2.1 Synthesis of Cobalt Selenide Nanoparticles

A series of cobalt selenide nanoparticles were synthesized at different reaction temperatures by 94 using one pot hydrothermal method. In a typical preparation, cobalt chloride hexa-hydrate 95 (713.8 mg, 3 mmol) was dissolved in 20 mL of deionized water under continuous stirring. 96 Selenium powder (236.0 mg, 3 mmol) was then slowly added into the above solution and 97 stirred for 1 hour at room temperature. Then hydrazine hydrate (8 mL) was added dropwise at 98 99 80 °C and the resulting suspension was kept on stirring at room temperature for one hour. The solution was then transferred into autoclave (100 mL Teflon container inserted in steel jacket) 100 followed by placing it into an oven at 180 °C for 16 hours. After cooling down the autoclave to 101 102 room temperature, the resulting product was thoroughly washed with deionized water and ethanol followed by centrifugation and vacuum drying at 90 °C. A similar synthetic strategy 103 was adopted to prepare cobalt selenide nanoparticles at different reaction times i.e. 4H, 8H, 104 12H and 20H. Schematic illustration of this procedure is shown in Figure S1. 105

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2.2 Electrode Fabrication

The electrode fabrication was done by drop-casting method on fluorine tin oxide (FTO) coated 107 glass. FTO coated glass was initially cleaned by sonicating in ethanol, then in ultrapure water 108 and finally in acetone for 15 minutes, followed by drying at 80 °C for 30 minutes. The ink was 109 prepared by ultra-sonicating 1 mg of electrocatalyst, 2 mL of ethanol, and two drops of binder 110 Nafion at 60°C. FTO coated glass was heated to 100 °C for 10 minutes and drop casting of ink 111 was made on pre-heated FTO glass using a micropipette to get uniformly coated working 112 electrode. The functional area was kept 1 cm² in each fabrication of the working electrode. 113 Finally, the coated electrodes were vacuum dried at 50 °C. 114

115 **3.** Characterizations

Powder X-ray diffraction (p-XRD) patterns of all prepared samples were recorded using 116 Bourevestnik Dron-8 diffractometer (St. Petersburg 190900, Russia) equipped with Cu-Ka 117 radiation ($\lambda = 1.5406$ Å) operated at 20 kV and 40 mA. For morphological and elemental 118 composition studies, scanning electron microscopy (SEM) imaging, Energy-dispersive X-ray 119 spectroscopy (EDX) and elemental mapping were performed with a Nova Nano-SEM 450 120 electrom microscope (Lincoln, NE 68588, United States). Electrochemical studies were 121 122 conducted on Gammry Interface 1000 electrochemical work station (Warminster, PA 18974, 123 United Staes) using conventional three-electrode system. Various electrochemical studies, including cyclic voltammetry (CV), linear sweep voltammetry (LSV), electrochemical 124 125 impedance spectroscopy (EIS), and Chronopotentiometry (CP) were conducted for each prepared material. These experiments were performed in a specialized electrochemical cell of 126 capacity 30 mL, and the electrolyte was 1 M KOH. 127

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4. Results and discussions

p-XRD analysis was conducted and the corresponding XRD patterns are shown in Figure 1. 129 130 The XRD patterns of 4H, 8H, 12H and 16H samples shows diffraction peaks at (100), (101), (102), (110), (201), (204) ⁴⁴ and these peaks were found to match with standard pattern of CoSe 131 (JCPDS-89-2004), whereas the XRD pattern of 20H sample was found matching with the 132 133 standard pattern of Co_{0.85}Se (JCPDS-52-1008). From 16H to 20H slight peak shift has been observed, it is assigned to Co_{0.85}Se hexagonal phase ⁵¹.All peaks were assigned to 134 corresponding crystal planes of both phases, and no extra peaks were detected, which shows 135 136 the high purity of products. But the relative intensities of peaks have been increased by increasing reaction time, due to difference in direction and alignment of planes in sample. It 137 has been observed that by decreasing reaction time from 20H to 4H size of nanoparticles has 138

been decreased due to which peak broadening has been occurred 52 . The reaction time longer than 16H generated vacancies in cobalt selenide, resulting in the formation of nonstoichiometric cobalt selenide, $Co_{0.85}Se$. The average crystallite sizes of the samples were calculated using the Debye–Scherrer formula and were found to be 34, 25, 17, 8 and 12 nm for 4H, 8H, 12H, 16H, and 20H samples, respectively. Moreover, the crystallinity index of asprepared materials was improved by increasing the reaction time.







Figure 1: p-XRD patterns of 4H, 8H, 12H, 16H and 20H

The morphology of as-prepared materials was analysed by scanning electron microscopy (SEM). As the catalytic reactions takes place on the surface so morphologies of as-synthesized materials are very important. CoSe nanoparticles exhibit dendrimeric structure and these dendrimers are expected to facilitate transportation of electron which will promote OER The

SEM micrographs of best performing catalyst, 16H, show the dendrimeric structure of 20-30 152 nanometer diameter, as shown in Figure 2 (a-c). In addition, open channel cavities of several 153 nanometer size can also be seen which are vital to enhance the mass diffusion process for 154 efficient kinetics. In case of 20H, fusion of dendrimeric structures into larger particles with a 155 156 diameter of 150-200 nm were observed, which reduce the active surface area of the catalyst (Figure S2 (d-f)). Aggregated micron-level sheet like structures can be seen in the images of 157 158 sample 12H (Figure S2 (a-c)). The elemental contents of 16H were determined by energy 159 dispersive X-ray (EDX) spectrometer, the atomic ratios of Co/Se were highly close to 1:1, which further validates the stoichiometric formation of CoSe. While in case of 20H, the 160 161 contents of cobalt were significantly lower than the selenide (atomic ratio of Co/Se was 0.82:1) which indicates the presence of Co vacancies in 20H, as observed in its p-XRD pattern. This 162 ratio is very close to empirical formula of $Co_{0.85}Se$ which further confirm its formation. The 163 EDX elemental mapping of 16H shows the homogeneous distributation of both Co and Se 164 throughout the matrix, as shown in Figure 2. 165



168 Figure 2 : (a)-(c) SEM images of dendrimer-like 16H; (d)-(f) Elemental mappings

Electrochemical performance of as-prepared materials for OER was evaluated in a three-170 electrode system, using standard calomel electrode and Pt as reference and counter electrode, 171 172 respectively, with prepared materials coated on FTO as working electrodes. The LSV was performed for different catalysts in 1M KOH solution at a scan rate of 5 mV s^{-1,} and the 173 corresponding recorded curves are shown in Figure 3(a). To obtain the 10 mAcm⁻² electron 174 density, the required overpotentials for 4H, 8H,12H, 16H and 20H were 360 mV, 290 mV, 280 175 mV, 250 mV, and 290 mV, respectively. All the prepared catalysts were found active for OER, 176 177 and the LSV for 16H sample initiated OER at remarkably low overpotential and reaching very 178 high current densities (570 mAcm⁻²) for oxygen evolution reaction under a small potential window. This high catalytic activity of 16H sample is due to its high crystallinity, porosity and 179

dendrimeric morphology compared to other prepared materials, as evident by p-XRD patterns and SEM micrographs. The samples with other preparation times (4H and 8H) showed the least peak electron densities and the highest overpotentials due to their more amorphous nature and compact sheet-like morphologies. CoSe synthesized at 12 h time interval (that is 12H) and the sample prepared in 20 h time interval (that is 20H) showed almost similar results for the water oxidation reaction. The lower catalytic activity of 20H as compared to 16H sample is due to its phase change and aggolomerated morphology despite having better crystallinity than the 16H.

187 The kinetics of catalyst was examined by Tafel plots, which shows the effect of overpotential or potential on constant current density. Tafel plots of all prepared CoSe samples were 188 189 investigated in the linear regions and fitted into the Tafel equation ($\eta = b \log J + a$) where b is Tafel slope. The resulting Tafel slopes for all prepared samples are shown in Figure 3(b). The 190 Tafel slope of 16H sample is 56 mV dec⁻¹, smaller than all other prepared samples and 191 192 benchmark RuO₂, implying an enhanced OER kinetics and improved bubble-releasing ability. These results suggest that high crystallinity, porous structure, dendrimeric morphology, and the 193 194 hexagonal CoSe phase improved the mass transfer and boosted the electron transfer process in water oxidation. These features make 16H a suitable candidate in practical applications with 195 more rapid OER rate. Linear sweep voltammograms were obtained at different scan rates of 5, 196 10, 25, 50, 75, 90 and 100 mVs⁻¹ for 16H sample to investigate the scan rate effect on the 197 current density and overpotential, as shown in Figure S3. The current density increased by 198 increasing the scan rate, demonstrating rapid kinetics on the active electrode materials and low 199 200 transport resistance between analyte and electrolyte for OER.

Electrochemical impedance spectroscopy (EIS) was conducted to understand better the kinetics
 of the OER process on prepared electrocatalysts. EIS was performed under a constant potential

203 of 0.5V and in the presence of 1.0 M KOH as an electrolyte. For each electrocatalytic experiment, a fresh electrolyte was used to minimize the differences in solution resistance. 204 Figures 3(c, d) show the respective Nyquist EIS plots and the bar graph for the charge transfer 205 206 resistances of all prepared electrocatalytic materials. A equivalent circuit was designed to fit 207 the EIS data, as shown in the inset of Figure 3(c). The designed circuit comprised of three resistors R1, R2 and R3 and two other elements Yo5,a5 and Yo7, a7, from this circuit R_s and 208 R_{ct} could be calculated. Solution resistance, R_s, is the resistance of the electrolyte, and R_{ct} is the 209 charge transfer resistance which is the interfacial resistance between the electrocatalyst and the 210 electrolyte ⁵⁰. As R_{ct} controls interfacial charge transfer kinetics which relates with the OER on 211 212 the surface of catalyst. So, smaller value of R_{ct} leads to efficient OER reaction. EIS plot and bar graph of different prepared materials show that the 16H represents the lowest charge 213 214 transfer resistance of just 0.218 Ω whereas the 4H showed highest charge transfer resistance of 5.07 Ω . The smallest charge transfer resistance for 16H among all other prepared materials 215 suggests rapid kinetics and better catalytic activity. This difference in charge transfer 216 217 resistances of as-synthesized materials shows that crystallinity, porosity and morphology play important roles in developing efficient electrocatalysts for water oxidation. 218

Electrochemically active surface area (ECSA) of all prepared materials was calculated using CV scans in the non-Faradic potential region at varied scanning rates (10 mV s^{-1} , 20 mV s^{-1} , 30 mV s^{-1} , 40 mV s^{-1} , 50 mV s^{-1}). A linear plot of scan rate versus capacitive current was derived from recorded CV data, and ther slope of this curve gave a quantitative value of double-layer capacitance (CdI). For these materials current linearly depends on the scan rate, which relates with capacittaive charging behaviour. The corresponding non-Faradic CV scans and linear plots for determination of CdI are shown in Figure S4. The magnitude of double-

layer capacitance is a measure of the ECSA. Abundant surface-active sites were found in 12H, 226 227 16H and 20H. The measured CdI values are 26.1, 20.4 and 17.5 mF cm⁻² for 12H, 16H and 20H, respectively. From these values of CdI, ECSA were estimated and are shown in the form 228 of bar graph in the Figure 3(e). The highest ECSA was observed for 16H followed by the 12H 229 230 and then for 20H. This trend indicates that materials' phase, crystallinity, morphology and porosity are critical in bringing accessible reaction sites for water oxidation. The sample 20H is 231 232 highly crystalline and porous, however, it has fewer active sites than 16H and 12H due to different CoSe phases and fusion of dendrimeric architectures. The results of ECSA suggests 233 that 16H sample had more exposed Co sites, leading to formation of more Co active species as 234 235 compared to other samples. Amorphous nature and micron-level sheet like structures leads to lowest ECSA for 4H followed by 8H. The ECSA results are in good agreement with other 236 characterizations and electrochemical evaluations. 237

238

Besides fast kinetics and higher catalytic activities, stability of electrocatalyst is another 239 240 essential criterion for commercial application. Chronopotentiometry of 16H sample was performed under constant current densities of 20 mA cm⁻² and 50 mAcm⁻² for over 24 hours. A 241 rich, continuous stream of oxygen bubbles were seen coming out of electrode surface during 242 the chronopotentiometric tests. The material 16H showed sustain potential values for OER 243 throughout the experiment, indicating its excellent stability under harsh oxidative conditions, 244 as shown in Figure 3(f). The percentage retention of electrocatalyst was above 95 percent, and 245 negligible increase in the overpotential was observed. A slight increase in voltage was 246 observed in both potentiometric experiments that could be a result of small removal of 247 electrocatalyst from the surface of electrode and decrease in contact between catalyst and 248

electrolyte due to continuous generation of oxygen gas. On the other hand, RuO₂ exhibit poor
stability with continuous attenuation of current density and chronoamperometric response of
RuO₂ was only 75 percent, as compared to initial current density under alkaline conditions.

252 These results predicted that dendrimeric cobalt selenide is outstanding catalyst for OER.

253 After the stability test, LSV of 16H was again conducted and compared with the LSV curve of fresh 16H. Negligible changes in the current density and overpotential were observed in both 254 255 LSV curves, as shown in the Figure S5. EIS of fresh 16H and after chronopotentiometric 256 stability test was also conducted. As shown in the Figure S6 and S7, a small increase in 257 resistance of 16H was observed after 24 hours stability test. These slight changes in current 258 density and resistance of 16H after stability experiment could be due to poisoning of very few active sites under highly oxidative conditions. In addition, powder XRD was also performed 259 260 before and after stability experiment and the resultant XRD patterns are shown in Figure S8. These patterns indicate that 16H ultimately retained its structure and the crystalline changes 261 were imperceptible, further confirming its excellent stability. 262

The performance of reported cobalt selenides has been illustrated in table S1 which shows that present work exhibits best results. The excellent catalytic activity of as-synthesized catalyst could be attributed to following factor: (i) dendrimeric structure which facilitated electron transport (ii) increased Co active sites

267 Possible mechanism of OER in alkaline conditions has been described below ⁵³:-

269
$$\operatorname{Co}(\operatorname{II}) + 3\operatorname{OH}^{-} \leftrightarrow \operatorname{Co}(\operatorname{III})\operatorname{OOH} + \operatorname{H}_{2}\operatorname{O} + e^{-}$$
 (1)

270
$$\operatorname{Co(III)OOH} + \operatorname{OH}^{-} \leftrightarrow \operatorname{Co(IV)O(OH)}_{2} + e^{-}$$
 (2)

271
$$\operatorname{Co(IV)O(OH)_2} + 2OH^- \leftrightarrow \operatorname{Co(IV)OO_2} + 2H_2O + 2e^-$$
 (3)

272
$$\operatorname{Co(IV)OO_2} + \operatorname{OH}^- \rightarrow \operatorname{Co(III)OOH} + \operatorname{O_2} + e^-$$
 (4)

273 Summary OER: $4OH^- \rightarrow O_2 + 2H_2O + 4e^-$

First three steps are reversible and explains entire OER process while fourth step is irreversible and fast. Co^{+2} firstly oxidized on the surface of electrode and forms CoOOH/CoSe, then it is oxidized to CoOO₂/CoSe, followed by its electrochemical oxidation and formation of O₂ and CoOOH. Moreover, dendrimeric structure of CoSe provides larger surfacea area and more active sites and enhance interfacial charge transfer which promotes OER reaction rate. Higher surface area also increase interface contact between electrolyte and electrode.

280





Figure 3 : Electrochemical studies of 4H, 8H, 12H, 16H and 20H (a) Linear sweep 284 voltammograms (b) Tafel plots in the linear regions (c) Nyquist EIS plots (d) charge transfer resistances (e) Electrochemically active surface areas (f) chronopotentiometric tests under 285 constant current densities of 20 mA cm⁻² and 50 mAcm⁻² for over 24 hours 286

In summary, we have synthesized dendrimeric nanoparticles of cobalt selenide at different 288 289 temperatures via hydrothermal method. It was observed that CoSe synthesized at 16H exhibits 290 excellent electrochemical activity with remarkably low overpotential (250 mV at 10 mA cm⁻²) and very high current density (570 mA cm⁻²) in a small potential window. The Tafel slope of 291 292 16H sample was 56 mV dec⁻¹ which indicates the faster kinetics at the catalyst surfaces. Moreover, it also showed excellent stability in harsh oxidative condition in a 24-hour long 293 stability test under alkaline conditions. Benefiting from higher stability, large electrochemically 294 active surface area, low overpotential and faster rate of electron transformation, cobalt selenide 295 electrocatalyst performance exceeds from commercial electrocatalysts. To the best of our 296 knowledge, the activity of as-synthesized catalyst is superior as compared to noble metal 297 catalysts. The remarkable OER efficiency is attributed to crystallinity, porosity, morphology and 298 nature of CoSe phase. It is important to note that crystalline and porous hexagonal CoSe is not 299 300 only economical but also very stable electrocatalyst and this work paves pathways for preparation of selenide based, efficient OER catalyst by introducing different transition metals 301 in it. 302

303 Credit authorship contribution statement

Muhammad Bilal: Experimental, Data curation, Writing original draft, Rashid: Data curation,
Validation, Amna Altaf: Data curation, Write up, Nadeem Baig: Data curation, Ghayoor Abbas
Chotana: Data curation, Raja Shahid Ashraf: Data curation, Shahid Rasool: Co-supervision,
Editing, Ayman Nafadi: Data curation; Manzar Sohail: Supervision, Conceptulization Writing
rereview & editing.

309	Declaration of	f competing interest
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The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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- Appendix A. Supplementary data Supplementary data to this article can be found online at

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